# Synthesis, Structural Parameter Determination and PL Characterization Sm<sup>3+</sup> Doped Mgb<sub>8</sub>o<sub>13</sub> Phosphor

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**Abstract:** A series of Sm doped  $MgB_8O_{13}$  phosphors were synthesized by solid state reaction method and their photo luminescence (PL) properties were systematically investigated. The synthesized materials were characterized using powder x-ray diffraction pattern (XRD) for confirmation. The emission and excitation of these materials were measured at room temperature by spectrofluorophotometer. The PL emission spectra of  $Sm^{3+}$  doped  $MgB_8O_{13}$  were observed at 565 nm and 607 nm in yellow and orange region respectively for 401 nm excitation near UV range. Emission at 565 nm and 607nm are assigned to  $4G_{5/2} \rightarrow {}^{6}H_{5/2}$  and  $4G_{5/2} \rightarrow 6H_{7/2}$  transition of  $Sm^{3+}$  ions, respectively. The effects  $Sm^{3+}$  concentration on the emission spectrum and luminescent intensity of  $MgB_8O_{13}$ : $Sm^{3+}$  were investigated. The photoluminescence properties of  $MgB_8O_{13}$ : $Sm^{3+}$  shows that the phosphor emits orange 607nm light and yellow 565nm under excitation 401 nm. Consequently, these materials are promising phosphors for white LEDs

Keywords: Photoluminescence (PL), XRD, Inorganic Borate; LEDs; Lamp phosphor.

#### 1. Introduction

In the past a few decades, rare earth borates have attracted considerable attention due to their practical applications as nonlinear optical (NLO), laser host, and luminescent materials. For example,  $YAI_3(BO_3)_4$ , RECa<sub>4</sub>O(BO<sub>3</sub>)<sub>3</sub> (RE= Y or Gd) [1], La<sub>2</sub>CaB<sub>10</sub>O<sub>19</sub> [2], andNa<sub>3</sub>La<sub>9</sub>O<sub>3</sub>(BO<sub>3</sub>)<sub>8</sub> [3] can be used as NLO crystals, Yb:YAI<sub>3</sub>(BO<sub>3</sub>)<sub>4</sub> andNd:GdCa<sub>4</sub>O(BO<sub>3</sub>)<sub>3</sub> [4] are potential self frequency doubling laser crystals, Rare earth ions have rich transition channels (f–f transitions and d–f transitions) and the characteristic of being not insensitive to lattice environment, long lifetime, and narrow spectra lines for f– f transitions, so they are widely used in laser and luminescent materials [5-8].

In order to search for new functional materials, considerable research should be carried out in the ternary systems like Li<sub>2</sub>O-BaO-B<sub>2</sub>O<sub>3</sub>. A suitable red-emitting phosphor for near-UV phosphor converted LEDs should have many potential applications, due to their excellent color rendering index, high color tolerance and high conversion efficiency into visible light [9]. Therefore, it is urgent to find new red phosphors that can be excited by NUV-LED chips for the fabrication of white LEDs. The f-f transition absorption and emission of the crystalline host activated by rare earths ions are of great importance due to their applications as luminescent optical materials emitting in the visible and near IR regions [10–12]. The selection of the rare earth ion as an activator is a key factor for the preparation of luminescence materials. Among the different rare earth ions, the  $Sm^{3+}$  ion as an activator is regarded as one of the most popular and efficient doping ions, which can produces intense orange light in the visible wavelength range. Sm<sup>3+</sup> ions in various hosts show bright emission in orange or red regions because of the transitions from the excited state  ${}^{4}G_{5/2}$  to the ground state  ${}^{6}H_{7/2}$  and also to the higher levels <sup>6</sup>H<sub>i</sub> (j=7/2, 9/2, and11/2) and found the application in high density optical storage, temperature sensors, undersea communications, various fluorescent devices, color display and visible solid-state lasers [13-14], Selection of suitable host material is also an important factor for the preparation of luminescent materials for different applications. Among the several inorganic host materials, the borate host matrix constitutes as one of the most important luminescence materials, which can produce plenty of crystal field environments imposed on emission contents [15, 16].On realizing the relevance and importance of these rare earth ions doped borate based phosphors, the present investigation is carried out by the authors. We have been synthesized  $Sm^{3+}$  ions doped MgB<sub>8</sub>O<sub>13</sub> host matrix by a conventional high temperature solid state reaction method in air atmosphere. Its structural, thermal, morphological and photoluminescence properties were systematically studied and also analyzed.

#### 2. Experimental

There are several low temperature methods such as hydrothermal and microwave methods which can be used instead of the conventional solid state reaction. But solid state synthesis is found to be more convenient than other methods due to economical interest and enhancement of the properties of the material. This paper describes the chemical composition of  $MgB_8O_{13}$ :Sm phosphor,host compound using other chemical element like Magnesium Carbonate ( $MgCO_3$ :99.9%), and Boric acid ( $H_3BO_3$ :99.9%) with rare earth elements like  $Sm_2O_3$ , etc., are used as starting materials. The stoichiometric amounts of the individual ingredients calculated on the basis of molar ratio are thoroughly blended and grounded thoroughly in an agate mortar with bricks and obtained a homogeneous mixture. This homogeneous mixture transferred into a chinacrucible which is then introduced into preheated muffle furnace maintained and sintered at 500°C for 3 hrs. The phosphor is then cooled to room temperature inside the closed furnace itself; subsequently, the china crucible is removed from the furnace and transferred the material in an mortar pestle for grounding for 30 minutes. The crushed powder is then transferred into silica crucible and is then introduced into preheated muffle furnace maintained and sintered at 800°C for 5 hrs. Again, the phosphors is then cooled to room temperature inside the closed furnace itself; subsequently, phosphors in silica crucible is then transferred in an morter pestle for grounding for 30 minutes. The closed furnace itself; subsequently, phosphors in silica crucible is then transferred in an morter pestle for grounding for 30 minutes.

Reaction:

 $MgCO_3 + 8H_3BO_3 + \frac{1}{2}Sm_2O_3 \rightarrow MgB_8O_{13}:Sm + CO_2 + 12H_2O$ 

The XRD technique was used in order to identify the product and check their crystallinity. The phase composition and phase structure were characterized by X-ray diffraction (XRD) pattern using a PANanalytical diffractometer with Cu K $\alpha$  radiation ( $\lambda$ =1.54 Å) operating at 40Kv, 30mA. The XRD data were collected in a 2 $\theta$  range from 10 to 80°, with the continuous scan mode. The photoluminescence (PL) emission spectra of the samples were measured at room temperature by using SHIMADZU Spectrofluorophotometer (RF-5301 PC) equipped with a 150W Xenon lamp as the excitation source. The same amount of sample was used in each case. Emission and excitation spectra were recorded using a spectral slit width of 3nm.

#### 3. Results and Discussion:

The structure of the samples were analyzed by an X-ray diffractometer (using Cu K $\alpha$  radiation with  $\lambda$ =1.54060(Å). The crystallite size (D) was calculated using the Debye Scherrer formula from the full-width at half-maximum (FWHM) ( $\beta$ 2 $\theta$ ).

$$D=0.9\lambda/(\beta_{(2\theta)}cos\theta)$$

Where  $\lambda$  is the wave length of X –rays used,  $\beta$  is the FWHM, and  $\theta$  is the angle between the incident and scattered X-rays.

The micro strain ( $\epsilon$ ) and the dislocation density ( $\rho$ ) of LiCaBO<sub>3</sub> material were evaluated using the equations

$$\varepsilon = \frac{\beta_{(2\theta)} \cos\theta}{4}$$
$$\rho = \frac{1}{D^2}$$

and

XRD pattern of  $MgB_8O_{13}$ :Sm phosphors results indicates that the final product was formed in homogeneous form. It is also observed that all the in different concentrations match with the standard peaks shown in figures 3.1 A,B,C,D.It indicates that samples are prepared correctly.The average size of the particle to be at 70.36569354 nm which proves that the prepared sample is nanomaterial. Mean dislocation density to be 0.002623- nm-2 and mean strain to be XRD pattern of 0.003526 phosphors result indicates that the final product shows crystalline nature.



Fig 3.1 (A) X-ray diffraction pattern of MgB<sub>8</sub>O<sub>13</sub> for without dopant.



Fig 3.1 (B) X-ray diffraction pattern of MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup>for 1mol%

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Fig 3.1 (C) X-ray diffraction pattern of MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> for 2 mol%



Fig 3.1 (D) X-ray diffraction pattern of MgB<sub>8</sub>O<sub>13</sub>:Sm for 3 mol%

| Sr. | 20     | d (obs.) | Height | FWHM  | D (nm)   | Strain  | Dislocation |
|-----|--------|----------|--------|-------|----------|---------|-------------|
| No. |        |          | (cps)  | (deg) |          |         | density     |
| 1.  | 14.071 | 6.289    | 46     | 0.52  | 15.40117 | 0.00904 | 0.004215    |
| 2.  | 16.495 | 5.370    | 63     | 0.125 | 63.7411  | 0.00184 | 0.0002461   |
| 3.  | 17.37  | 5.10     | 9      | 1.1   | 7.3095   | 0.00153 | 0.01871     |
| 4.  | 19.44  | 4.562    | 33     | 0.98  | 8.2574   | 0.00121 | 0.01466     |
| 5.  | 21.829 | 4.068    | 50     | 0.42  | 19.3407  | 0.00457 | 0.002673    |
| 6.  | 24.38  | 3.698    | 39     | 1.27  | 6.4253   | 0.00122 | 0.02422     |
| 7.  | 27.38  | 3.255    | 66     | 1.33  | 6.1724   | 0.00111 | 0.026247    |
| 8.  | 31.10  | 2.873    | 33     | 0.72  | 11.4989  | 0.00520 | 0.007562    |
| 9.  | 32.85  | 2.724    | 21     | 0.60  | 13.8592  | 0.00405 | 0.005206    |
| 10. | 35.86  | 2.502    | 13     | 0.66  | 12.7020  | 0.00398 | 0.006198    |
| 11. | 40.15  | 2.2444   | 21     | 0.40  | 21.2303  | 0.00206 | 0.002218    |
| 12. | 44.26  | 2.045    | 7      | 2.5   | 3.4442   | 0.00111 | 0.084299    |
| 13. | 50.01  | 1.8223   | 7      | 0.21  | 41.9099  | 0.00768 | 0.000569    |
| 14. | 54.09  | 1.6940   | 8      | 0.16  | 55.9716  | 0.00505 | 0.0003192   |
| 15. | 55.93  | 1.6427   | 10     | 0.11  | 82.0968  | 0.00324 | 0.0001483   |

Table 3.2.Structure parameters of intrinsic MgB<sub>8</sub>O<sub>13</sub>

3.2 Photoluminescence (PL) characterization of MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> phosphor



Fig 3.2(A) : PL characterization of MgB<sub>8</sub>O<sub>13</sub>Sm<sup>3+</sup>Phosphor



Fig3.2(B) PL emission spectra of MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> monitored at 401nm excitation



Fig 3.2(C) : PL excitation spectra of MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> monitored at 607nm emission

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| Concentration                             | Excitation      | Emission        | Intensity (a.u) |
|---|-----------------|-----------------|-----------------|
|   | Wavelength (nm) | wavelength (nm) |                 |
| MgB <sub>8</sub> O <sub>13</sub> :Sm (1%) | 401.49          | 607.60nm        | 98              |
| MgB <sub>8</sub> O <sub>13</sub> :Sm (2%) | 401.83          | 606.78 nm       | 99              |
| MgB <sub>8</sub> O <sub>13</sub> :Sm (3%) | 402.83          | 606.96 nm       | 105             |
| MgB <sub>8</sub> O <sub>13</sub> :Sm (4%) | 401.49          | 606.37 nm       | 87              |

Table 3.3: Variation in emission intensity of MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> with concentration of Sm





To study photoluminescence characteristics the excitation spectra was recorded for MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> phosphor by monitoring emission at 607 nm as depicted in Fig3.2(C). The excitation spectra MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> phosphor shows dominant. peak at 401 nm due to  ${}^{6}H_{5/2} \rightarrow {}^{4}F_{7/2}$  transition Upon this excitation the photoluminescence emission spectra for different concentration of Sm<sup>3+</sup> ion were recorded in the spectral range 550-750 nm shown in Fig 3.2(B). In the emission spectra of MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> the peak is located at 607 nm which is due to  ${}^{4}G_{5/2} \rightarrow {}^{6}H_{7/2}$  and other peak is located at 565 nm which is due to  ${}^{4}G_{5/2} \rightarrow {}^{6}H_{5/2}$ . It shows the significant orangered emission at 607 nm and yellow emission at 565nm. It is found that the shape and profile for the emission spectra do not vary with the change of the Sm<sup>3+</sup> doping concentration, but the PL intensity

changes greatly . The optimum concentration was determined to be 3 mole % to obtain strongest PL emission intensity.

# 4. Conclusion:

All the XRD patterns of  $MgB_8O_{13}$ :Sm<sup>3+</sup> phosphors indicate that the final product was formed in homogeneousform. The average size of the particle to be at 24.62403 nm which proves that the prepared sample is nanomaterial. Mean dislocation density to be 0.013166 nm-2 and mean strain to be 0.003526. Results of all theXRD patterns of  $MgB_8O_{13}$ :Sm<sup>3+</sup> phosphors indicates that the final products are incrystalline form.

 $Sm^{3+}$  doped MgB<sub>8</sub>O<sub>13</sub> phosphor have been synthesized by high temperature solid state reaction method. The photoluminescence properties of MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> shows that the phosphor emitsyellow (565nm) and orange red(607nm) light under excitation of 401 nm wavelength. The result indicates that MgB<sub>8</sub>O<sub>13</sub>:Sm<sup>3+</sup> is an efficientyellow and orange red emitting phosphors and is suitable material for UV LEDs.

# Acknowledgements

Author VRK is thankful to management of the Institution KNM, Nagpur for providing useful facilities of the instrumentation, SHIMADZU Spectrofluorophotometer (RF-5301 PC) to carry out this work.

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